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1 **Gelatin- and starch-based hydrogels. Part A: Hydrogel development,**
2 **characterization and coating.**
3

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23 **Abstract**

24 The present work aims at constructing the ideal scaffold matrix of which the physico-chemical
25 properties can be altered according to the targeted tissue regeneration application. Ideally, this
26 scaffold should resemble the natural extracellular matrix (ECM) as close as possible both in
27 terms of chemical composition and mechanical properties. Therefore, hydrogel films were
28 developed consisting of methacrylamide-modified gelatin and starch-pentenoate building blocks
29 because the ECM can be considered as a crosslinked hydrogel network consisting of both
30 polysaccharides and structural, signaling and cell-adhesive proteins. For the gelatin hydrogels,
31 three different substitution degrees were evaluated including 31%, 72% and 95%. A substitution
32 degree of 32% was applied for the starch-pentenoate building block. Pure gelatin hydrogels films
33 as well as interpenetrating networks with gelatin and starch were developed. Subsequently, these
34 films were characterized using gel fraction and swelling experiments, high resolution-magic
35 angle spinning ¹H NMR spectroscopy, rheology, infrared mapping and atomic force microscopy.
36 The results indicate that both the mechanical properties and the swelling extent of the developed
37 hydrogel films can be controlled by varying the chemical composition and the degree of
38 substitution of the methacrylamide-modified gelatin applied. The storage moduli of the
39 developed materials ranged between 14 and 63 kPa. Phase separation was observed for the IPNs
40 for which separated starch domains could be distinguished located in the surrounding gelatin
41 matrix. Furthermore, we evaluated the affinity of aggrecan for gelatin by atomic force
42 microscopy and radiolabeling experiments. We found that aggrecan can be applied as a bioactive
43 coating for gelatin hydrogels by a straightforward physisorption procedure. Thus, we achieved
44 distinct fine-tuning of the physico-chemical properties of these hydrogels which render them
45 promising candidates for tissue engineering approaches.

46
47

48 **Key words:** gelatin, starch, biomaterials, aggrecan, tissue engineering

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49 **1. Introduction**

50 The lack of acutely available organs for transplantation is a worldwide issue which is even
51 expected to worsen as the world population ages. Tissue engineering is an approach aiming at
52 bridging this gap.(Furth, Atala, & Van Dyke, 2007; Griffith & Naughton, 2002; Langer R, 1993;
53 Langer, 1997; Lemons, 2013) In this approach, cells are seeded onto scaffolds or implants to
54 develop into functional tissues.(Drury & Mooney, 2003; Gomillion & Burg, 2006; C. Liu, Xia, &
55 Czernuszka, 2007; Lutolf & Hubbell, 2005; Peters et al., 2009) In addition, an increasing number
56 of procedures can be found in literature which rely on the application of stem cells.(Barry &
57 Murphy, 2004; Gomillion & Burg, 2006; Griffith & Naughton, 2002; Jeffrey M. Gimble et al.,
58 2007; Peters et al., 2009) Using mesenchymal stem cells (MSC), the present study aims at a
59 scaffold guided strategy towards tissue regeneration. The constructed scaffold is a three-
60 dimensional matrix serving as a surrogate extracellular matrix (ECM) enabling cell attachment
61 and promoting cell proliferation as well as differentiation. The design of a scaffold resembling
62 the natural ECM is preferred in order to mimic as closely as possible the natural aqueous
63 environment that cells are experiencing.(Chen, Wang, Wei, Mo, & Cui, 2010; Kim, Kim, &
64 Salih, 2005; Kuo, Chen, Hsiao, & Chen, 2015) This natural ECM can be considered as a
65 crosslinked hydrogel network consisting of polysaccharides as well as structural, signaling and
66 cell-adhesive proteins. Taking this knowledge into consideration, it is of great interest to evaluate
67 the potential of polymer networks mimicking this ECM composition. Therefore, gelatin and
68 starch are applied as natural building blocks in the present work, representing both the protein
69 and polysaccharide constituent of the natural ECM.

70 Gelatin is derived from collagen, which is the most abundant structural protein in mammals.(Di
71 Lullo, Sweeney, Korkko, Ala-Kokko, & San Antonio, 2002) In addition, it is generally non-
72 immunogenic and retains informational signals including an arginine-glycine-aspartic acid
73 (RGD) sequence which promotes cell adhesion, differentiation and proliferation.(Gautam, Dinda,
74 & Mishra, 2013) These properties and its unique gel-forming ability render gelatin an interesting
75 biopolymer towards tissue engineering applications.(Awad, Quinn Wickham, Leddy, Gimble, &
76 Guilak, 2004; Dubruel et al., 2007; Li et al., 2005; Nichol et al., 2010) Starch, on the other hand,
77 consists of a mixture of the polysaccharides amylose and amylopectin. The relative ratio of
78 amylose to amylopectin strongly depends on the starch source considered. The application of
79 starch offers several advantages including its biodegradability and ease of processing.(Azevedo,
80 Gama, & Reis, 2003; Puppi, Chiellini, Piras, & Chiellini, 2010) Starch-based polymers as well as
81 blends have already been introduced as promising biomaterials for bone and cartilage tissue
82 engineering applications due to these advantages. For instance, Mendes et al. (2001) showed the
83 potential of starch/ethylene vinyl alcohol blends reinforced with hydroxyapatite for temporary
84 bone replacement implants.(Mendes et al., 2001) Raafat et al. (2013) developed a hydrogel series
85 composed of starch/N-vinylpyrrolidone which were proven to exhibit *in vitro* bioactivity and
86 blood compatibility.(Raafat, Eldin, Salama, & Ali, 2013) Moreover, gelatin and starch are often

87 combined for several food processing applications.(Burey, Bhandari, Rutgers, Halley, & Torley,
88 2009; Firoozmand, Murray, & Dickinson, 2009; MARRS, 1982)

89 In this work, hydrogels were developed consisting of either a gelatin phase or the combination of
90 both a starch and a gelatin phase. In the latter case, these hydrogels are so-called interpenetrating
91 polymer networks (IPNs) if the appropriate crosslinking strategy is applied ensuring both
92 building blocks to be covalently crosslinked but not bonded to each other.(V et al., 2007) The
93 potential of gelatin hydrogels in contact with adipose tissue derived mesenchymal stem cells
94 (adMSCs) was already demonstrated by Peters et al. (2009) towards the adhesion of these
95 cells.(Peters et al., 2009) Therefore, we selected the gelatin hydrogels as reference material for
96 the IPNs of starch and gelatin. Pure starch hydrogels were not applied as these hydrogels were
97 shown to be too brittle to process them in hydrogel films. To the best of our knowledge, we first
98 reported on the combination of starch and gelatin in IPNs for the purpose of tissue engineering
99 applications. Indeed, previous results reported by Van Nieuwenhove et al. (2015) on starch-based
100 hydrogels were promising since the hydrogels developed in contact with adMSC were shown to
101 be biocompatible.(Van Nieuwenhove et al., 2015)

102 IPNs have gained an increased attention the last decades mainly due to their high potential as
103 hydrogels for biomedical applications.(Dragan, 2014) However, most of the hybrid IPNs
104 hydrogels, reported in literature, are obtained by either combining various polysaccharides or
105 synthetic polymers and proteins with synthetic polymers.(Dragan, 2014; La Gatta, Schiraldi,
106 Esposito, D'Agostino, & De Rosa, 2009; Peng, Yu, Mi, & Shyu, 2006; Pescosolido et al., 2011)
107 Only a few papers report on the combination of proteins and polysaccharides for the construction
108 of (semi)-IPNs.(Cui, Jia, Guo, Liu, & Zhu, 2014; Y. Liu & Chan-Park, 2009; Picard, Doumèche,
109 Panouillé, & Larreta-Garde, 2010; Turgeon & Beaulieu, 2001)

110 The present work focusses on the construction of the ideal scaffold matrix of which the physico-
111 chemical properties can be altered according to the targeted tissue regeneration application. The
112 latter is highly relevant as natural tissue is also characterized by different mechanical properties.
113 Thus, altering the mechanical properties of the constructed hydrogel films is of great interest. For
114 instance breast tissue, mainly composed of adipose tissue, is characterized by a storage modulus
115 of 3.2 kPa(Abbas, Judit, & Donald, 2007), whereas the storage modulus of articular cartilage is in
116 the range of 2 to 7 GPa(Silver, Bradica, & Tria, 2002). Due to their soft and rubbery consistence,
117 hydrogels do not reveal such high storage moduli. However, these hydrogels can still be
118 applicable as coating onto implants to target orthopedic applications.

119 For this reason, hydrogel films were prepared with varying chemical composition (i.e. ratio
120 between gelatin and starch phase) and varying degree of substitution (DS) of the gelatin phase
121 applied. First, gelatin and starch were chemically modified with photo-crosslinkable moieties.
122 This modification enables their subsequent processing into hydrogel films and ensures sufficient
123 stability of the materials upon *in vitro* application. In addition, the present work will evaluate
124 whether a bioactive coating of aggrecan, the main articular cartilage constituent, can be deposited

125 onto the materials via physisorption. More specifically, liquid atomic force microscopy and
126 radiolabeling experiments will be performed to study this hydrogel coating.

127

128

129 **2. Experimental section**

130 **2.1. Materials**

131 For all the synthesis experiments, gelatin (type B), from bovine bone origine, was applied
132 (Rousselot, Gent, Belgium). Furthermore, dimethyl sulfoxide (DMSO, 99.85%) was purchased
133 from Acros (Geel, Belgium) and purified via distillation before use. Irgacure® 2959 was applied
134 as photo-initiator (BASF, Kaisten, Switzerland) and dithiothreitol (Fisher Scientific,
135 Erembodegem, Belgium) was used as a bifunctional thiol-based crosslinker agent. All other
136 chemicals were purchased from Sigma Aldrich (Bornem, Belgium) and were used as received
137 unless stated otherwise. The radiolabeling experiments were performed using Iodogen (1,3,4,6-
138 tetrachloro-3a,6a-diphenyl-glycouril) obtained from Pierce (USA) and using a radioiodide
139 solution (125I: Perkin Elmer, Massachusetts, USA).

140

141 **2.2. Synthesis of hydrogel building blocks**

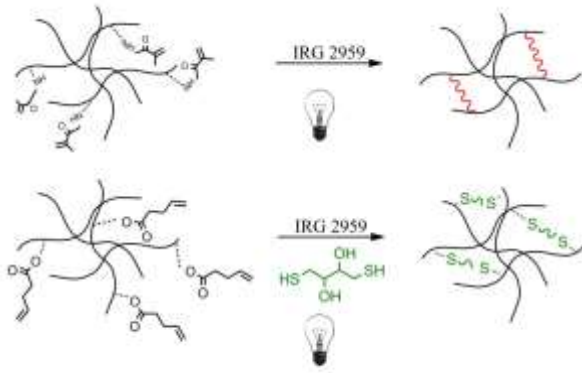
142 Both the pentenoate-modified starch (SP) and the methacrylamide-modified gelatin
143 (gel-MA) were synthesized as described earlier.(Peters et al., 2009; Van Nieuwenhove et al.,
144 2015) In brief, corn starch was dissolved in DMSO (5 w/v%, 70 °C), a catalytic amount of
145 dimethylaminopyridine was added and the reaction mixture was stirred for 20 minutes.
146 Subsequently, 4-pentenoic anhydride (37.5 equivalents with respect to the saccharide units) was
147 added and reacted overnight. The purified product was obtained via precipitation in ethanol,
148 followed by dialysis against double distilled water (MWCO: 12,000 – 14,000 Da) and freeze-
149 drying by means of a Christ freeze-dryer alpha 2-4-LSC.

150 For the gelatin derivatives, the amount of crosslinkable side chains was adjusted by varying the
151 amount of methacrylic anhydride added. Three different modifications were performed using 0.5,
152 1 and 2.5 equivalents methacrylic anhydride added with respect to the primary amines present
153 along the gelatin backbone.

154 **2.3. Hydrogel production**

155 Hydrogel films were prepared through covalent crosslinking. For this purpose, a gel-MA solution
156 (10 w/v%) was crosslinked via photo-induced polymerization in the presence of 2 mol%
157 Irgacure® 2959 upon applying UV-A irradiation for 30 minutes (with an intensity of 10 mW/cm²
158 and a wavelength range of 250-450 nm). IPNs were obtained by the addition of one equivalent of
159 DTT and Irgacure® 2959 to various SP (5 w/v%) and gel-MA solutions (10 w/v%) which were

160 subsequently exposed to UV-A irradiation. The addition of DTT is needed as the crosslinking of
 161 SP occurred via a radical thiol-ene reaction.
 162



163
 164 **Figure 1** UV-Crosslinking of methacrylamide-modified gelatin solution (top) upon the addition of Irgacure 2959® and starch-
 165 pentenoate (below) upon the addition of Irgacure 2959® and a bifunctional thiolcrosslinker.

166

167 2.4. Characterization of the hydrogels developed

168

169 2.4.1. Gel fraction and swelling experiments

170 Samples ($\phi = 1.4$ mm, thickness = 1 mm) of the crosslinked hydrogels were incubated in double-
 171 distilled water at 37°C in order to determine the gel fraction of the crosslinked hydrogels. As a
 172 result, polymer chains that were not covalently linked into the network were able to leach out
 173 from the hydrogels by diffusion. The gel fraction can be calculated, expressed as the percentage
 174 of material which is chemically incorporated in the three-dimensional network (equation 1).

175

$$176 \text{ gel fraction (\%)} = \frac{W_d}{W_{d0}} \cdot 100 \quad (1)$$

177

178 with W_d = dry weight after swelling

179

180 W_{d0} = dry weight before swelling

181

182 All the measurements were performed in duplicate. The results are presented as mean values with
 183 corresponding standard deviations (SD).

184

185 For the swelling experiments, the hydrogel films were submerged in double-distilled water at
 37°C, and the changes in mass were recorded as a function of time. At distinct time points, the

186 samples were removed from the medium, dipped on a piece of paper in order to remove adhered
187 solution to the surface, and weighed. Afterwards, the samples were again incubated in the
188 swelling medium.

189

190 The swelling percentage can be defined as:

191

$$192 \quad \text{Swelling (\%)} = \frac{W_{ht} - W_{do}}{W_{do}} \cdot 100 \quad (2)$$

193 with W_{do} = weight of dry gel at initial time 0

194 W_{ht} = weight of hydrated gel at time t

195

196 All these experiments were performed in duplicate. The results are reported as mean values with
197 corresponding SD.

198

199

200 **2.4.2. Determination of crosslinking efficiency via HR-MAS ¹H-NMR spectroscopy**

201 High Resolution Magic Angle Spinning ¹H NMR spectroscopy (HR-MAS) was performed in
202 order to evaluate the crosslinking efficiency (CE) of the developed hydrogel films. A Bruker
203 Avance II 700 spectrometer (700.13 MHz) device was used applying a HR-MAS probe equipped
204 with a ¹H, ¹³C, ¹¹⁹Sn and gradient channel. The spinning rate was adjusted to 6 kHz.

205 On the day of the experiments, a small amount of the freeze-dried hydrogels was placed inside a
206 4 mm zirconium oxide MAS rotor (50 μl) and a few microliters of deuterium oxide (D₂O) were
207 added enabling the samples to swell. A teflon-coated cap was applied in order to close the rotor.
208 Prior to analysis the HR-MAS samples were homogenized by manual stirring. Afterwards, the
209 spectra were analyzed after baseline correction.

210 The CE is calculated using the following equation (Sandra Van Vlierberghe José C. Martins, and
211 Peter Dubruel, 2010):

212

$$213 \quad CE (\%) = \left[\frac{\left(\frac{I_{5.75 \text{ or } 5.1 \text{ ppm}}^k}{I_{1.1 \text{ ppm}}^k} \right) - \left(\frac{I_{5.75 \text{ or } 5.1 \text{ ppm}}^c}{I_{1.1 \text{ ppm}}^c} \right)}{\left(\frac{I_{5.75 \text{ or } 5.1 \text{ ppm}}^k}{I_{1.1 \text{ ppm}}^k} \right)} \right] \times 100 \quad (3)$$

214

215 This equation (3) is based on the comparison of the intensity of the signals characterizing the
216 protons of the introduced double bonds, before and after crosslinking. Normalization is applied
217 by using the inert signal at 1.1 ppm, because different samples need to be compared.

218

219 **2.4.3. Rheology**

220 The mechanical properties of the hydrogels were investigated via oscillation rheology with a
221 rheometer type Physica MCR-301 (Anton Paar, Sint-Martens-Latem, Belgium) running with
222 Physica Rheoplus software. All measurements were performed using a plate-plate geometry.
223 More specifically, a hydrogel sample was placed between two parallel plates (diameter upper
224 plate = 25 mm), after which the upper plate was adjusted to ensure close contact of each sample
225 with both plates. Tests were performed using oscillatory sine functions and upon applying a
226 frequency of 1 Hz and a gap setting of 0.95 mm. In addition, a 0.05% strain was selected to
227 perform the oscillatory measurements as the linear visco-elastic range ranges from 0 to about
228 0.3% strain (data not shown). In the present work, the different hydrogels were measured under
229 these settings while monitoring the storage (G') and the loss moduli (G'').

231 **2.4.4. Atomic Force Microscopy and IR-mapping**

232 Atomic force microscopy (AFM) experiments were performed with a Nanoscope IIIa Multimode
233 (Digital Instruments, Santa Barbara, California, USA) applying 'tapping mode' in air.
234 Measurements were performed on spincoated gelatin/starch solutions (10 w/v % gelatin and 5
235 w/v% starch solution) since AFM measurements require flat surfaces. In addition, spincoated
236 gelatin and starch solutions were also measured separately as references. The nanoscope software
237 version 4.43r8 was used to process all data obtained with AFM. On the other hand, IR-mapping
238 was performed on dried hydrogel films using a Perkin Elmer Spectrum 100 FT-IR spectrometer
239 with a Spotlight 400 FT-IR imaging system. Therefore, the hydrogel surfaces were scanned using
240 IR mapping to evaluate the absorbance potentially occurring at the characteristic wavenumbers
241 for gelatin and starch in order to determine the presence of both building blocks in the hydrogel
242 samples."

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246 **2.5. Characterization of bioactive coating**

248 In the present work, AFM and radiolabeling experiments were utilized in order to determine the
249 interaction between gelatin and aggrecan.

250 **2.5.1. AFM under liquid conditions**

251 AFM experiments were conducted on an Agilent 5500 AFM/SPM microscope in a liquid
252 environment at 20 °C.

253 **2.5.1.1. Topographic AFM imaging**

254 Prior to AFM imaging, aggrecan from bovine plasma was dissolved in phosphate buffered saline
255 (PBS) to acquire a stock solution of 1 mg/ml. The aggrecan solution was diluted to the desired

256 concentration and added onto the gelatin hydrogel film for 30 minutes at room temperature
257 followed by three PBS washing steps prior to imaging. The washing steps were essential to
258 remove loosely bound aggrecan.

259 Images were obtained in tapping mode using silicon tips (Nanosensors, series PPP-NCSTR-50)
260 with a resonance frequency within a range from 76 to 263 kHz and a force constant of 12 – 29
261 N/m. Typical scan rates were in the range of 0.5 – 1 kHz at a resolution of 512 points/line.
262 All measurements were performed in PBS.

263 **2.5.1.2. Force spectroscopy**

264 Force spectroscopy measurements were performed using a backside aluminium coated silicon
265 cantilever (Cont GB-G, Budget Sensor) with a nominal spring constant of 0.02 N/m and a
266 resonant frequency of 13 kHz. Accurate measurement of spring constants was obtained using the
267 equipartition theorem (Thermal K)(Hutter & Bechhoefer, 1993). Forces of interaction between
268 the aggrecan and the hydrogel were measured by functionalizing the AFM tip with aggrecan
269 through a physisorption process by incubation of the tip for 30 minutes. Prior to monitoring the
270 aggrecan interactions with the gel, force distance curves were acquired on bare mica in order to
271 confirm that the tip was successfully functionalized. Force spectroscopy experiments were
272 performed on the gelatin samples at four locations defined by the user. Approximately 1000 –
273 1500 force-distance curves were obtained per location.

274 For the analysis of the data obtained, Scanning Probe Image Processor (SPIP) version 6.2.8
275 (Image Metrology, Lyngby, Denmark) was used. Interaction forces between the aggrecan and the
276 gel were derived from the registered force distance curves. Histograms of the height features as
277 well as the rupture forces were created with Sigmaplot (Systat Software, San Jose, CA). For the
278 rupture force distributions of aggrecan, the selected curves were fitted to a Gaussian function in
279 order to extract the average rupture force.

280

281 **2.5.2. Radiolabeling experiments**

282 Radioiodination was performed by a slightly modified method described by Pierce Biotechnology
283 Inc. (Rockford, IL, USA; www.piercenet.com). In brief: Iodogen was dissolved in chloroform to
284 a concentration of 2 mg/mL and 100 μ L was added to a 5 mL conical vial. The solvent was then
285 evaporated under a gentle N₂ flow at room temperature and the Iodogen-coated vials were stored
286 in a dessicator at 5 °C prior to use. A stock solution of aggrecan (0.5 mg/ml, 1.5 ml) was added to
287 a Iodogen coated reaction vessel, immediately followed by the addition of 20 μ L radioiodide
288 solution (¹²⁵I). This mixture was incubated for 20 minutes at room temperature under slight
289 shaking. Free iodine was removed by G-25 Sephadex gel filtration (GE Healthcare, Belgium),
290 equilibrated with 0.01 M phosphate buffer of pH 7. The overall radiochemical purity (RCP) was
291 then determined using iTLC-SG chromatographic strips (Gel- man Sciences) and a citrate-buffer

292 (0.068 M citrate, pH 7.4) as eluent. From this ¹²⁵I-aggrecan solution dilutions were prepared to
293 adjust the concentration of aggrecan to 0.5, 0.3, 0.2, 0.1 and 0.05 mg/mL. The procedure for
294 coating the hydrogel films is similar to the aforementioned in section 1.5.1.1.

295

296 **3. Results and discussion**

297 **3.1 In-depth physico-chemical characterization of the hydrogels**

298 Gelatin and starch were modified with UV-crosslinkable side-groups enabling their subsequent
299 processing into hydrogel films. Gelatin was successfully modified with varying amount of
300 methacrylic anhydride.(Peters et al., 2009; Salamon et al., 2014) In this way, the influence of the
301 DS on the mechanical properties could be evaluated. The modification was confirmed and
302 quantified via ¹H-NMR spectroscopy for the different gelatin derivatives (see figure S1). The
303 methacrylamide-modified gelatins
304 (gel-MA) in the present work possess a DS of 31, 72 and 95% with respect to the primary amines
305 available along the gelatin backbone. In addition to the functionalized gelatin, starch was
306 successfully modified using 4-pentenoic anhydride yielding starch-pentenoate (SP) with a DS of
307 32%.(Van Nieuwenhove et al., 2015) This DS was also quantified by means of ¹H-NMR
308 spectroscopy and is expressed as the amount of modified repeating saccharide units (see figure
309 S1).

310 Subsequently, hydrogel films of both gel-MA and gel-MA in combination with SP were prepared
311 via film casting followed by chemical crosslinking. This enabled the characterization of the
312 developed materials via several techniques. Pure starch hydrogels were not developed as these
313 hydrogels were not robust enough to enable manipulation.

314

315 **3.1.1. Gel fraction and swelling experiments**

316

317 First, the gel fractions and the equilibrium swelling degree of the developed materials were
318 determined. The results are listed in table 1 and to facilitate further discussion each hydrogel
319 sample is designated with a unique code. On the one hand, gel-MA x% indicates hydrogels
320 purely based on gelatin which are characterized by their DS represented by x%. On the other
321 hand, the abbreviation SP1 reflects the presence of a SP content of 10% and SP2 assigns the IPNs
322 defined by 20% SP content. The gel fraction results indicate an efficient crosslinking during
323 which most of the crosslinkable moieties were consumed resulting in gel fractions of 85% and
324 higher. In general, thus, well-established networks were formed as almost no leaching occurred
325 of unbound molecules. As anticipated, it can be observed in table 1 that the gel fraction will
326 increase with an increasing DS for gelatin hydrogels. This because a higher DS will result in a
327 more crosslinked hydrogel network going from gel-MA 31% to 95%.

328

329 A small decrease in the gel fraction can be observed for the hydrogel samples with a starch
330 content of 20% compared to 10% and the hydrogels without starch. However, conversely, an

331 increase in total amount of crosslinkable moieties is noticeable with an increasing amount of
332 starch present in the polymer network (see last column table 1). It can be hypothesized that upon
333 the introduction of a critical amount of starch (i.e. 20%), the phase separation between starch and
334 gelatin will be more pronounced causing the starch to be more clustered together in domains. The
335 occurrence of phase separation will still be tackled in depth in section 2.1.4.

336
337 Therefore, it is hypothesized that upon introducing this critical amount of starch the gel fraction
338 again decreases because the starch domains leach out during incubation (cfr. these domains can
339 be considered as starch-only hydrogels, which do not enable manipulation as already indicated
340 above). This effect is not demonstrated in the hydrogel samples with a 10% starch content, since
341 the starch hydrogel building blocks will be more randomly distributed in a gelatin phase.

342
343 The swelling experiments show that all hydrogel types are able to absorb large quantities of
344 water. Indeed, equilibrium swelling degrees ranging from 660% up to 4100% were observed for
345 the hydrogel samples developed. These results are in good agreement with the results obtained by
346 Graulus et al. (2015) for gelatin hydrogels and hydrogels consisting of gelatin and
347 alginate.(Graulus et al., 2015)

348

349 **Table 1** Gel fractions (%) for the various hydrogel samples and the number of crosslinkable moieties present in the precursor solutions (methacrylamide for gel-MA
 350 versus pentenoate for the starch phase). All measurements were performed in duplicate and the results are presented as mean values with corresponding standard
 351 deviations (SD) (n=2).

Code	Composition (v%)		Gel fraction (%) ± SD	mol MA moieties / ml precursor solution	mol pentenoate moieties / ml precursor solution	mol total amount of crosslinkable moieties/ ml precursor solution
	gel-MA	SP				
gel-MA 31%	100	-	85 ± 5	1.19E-05	-	1.19E-05
gel-MA 72%	100	-	94 ± 1	2.77E-05	-	2.77E-05
gel-MA 72% - SP1	90	10	100 ± 1	2.49E-05	9.87E-06	3.48E-05
gel-MA 72% - SP2	80	20	86 ± 4	2.22E-05	1.97E-05	4.19E-05
gel-MA 95%	100	-	98 ± 1	3.66E-05	-	3.66E-05
gel-MA 95% - SP1	90	10	100 ± 9	3.29E-05	9.87E-06	4.28E-05
gel-MA 95% - SP2	80	20	93 ± 7	2.93E-05	1.97E-05	4.90E-05

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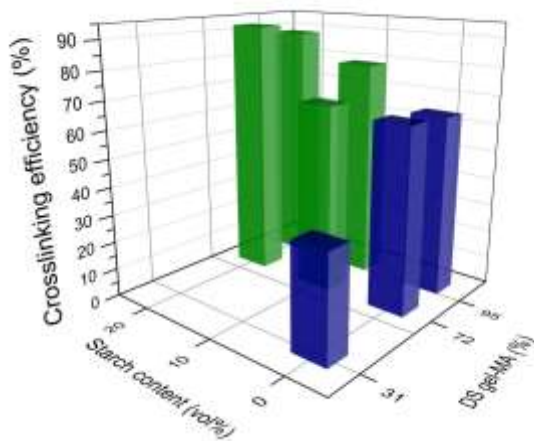
358

359 **3.1.2. Evaluation of crosslinking efficiency**

360
361 The crosslink efficiency (CE) of the UV-cured hydrogels was evaluated by means of HR-MAS
362 ¹H-NMR spectroscopy. This technique evaluates the consumption of double bonds upon
363 crosslinking and is thus a measure for the efficiency of crosslinking.(Sandra Van
364 Vlierberghe José C. Martins, and Peter Dubruel, 2010) Conventional ¹H-NMR spectroscopy
365 does not enable the characterization of crosslinked polymer networks due to the considerable line
366 broadening which results from the presence of dipolar couplings and magnetic susceptibility
367 effects.(Ramadhar, Amador, Ditty, & Power, 2008; Shapiro, Chin, Marti, & Jarosinski, 1997)
368 HR-MAS spectroscopy circumvents this line broadening by rapidly rotating the sample at a
369 magic angle of 54.7° with respect to the static magnetic field, following swelling of the
370 material.(Ramadhar et al., 2008) This swelling induces sufficient, solution-like, rotational
371 mobility of the polymer.(Sandra Van Vlierberghe José C. Martins, and Peter Dubruel, 2010)
372 Highly crosslinked hydrogel materials will thus exhibit a reduced chain mobility and will show
373 broader peaks compared to less crosslinked materials.(Rueda, Suica, Komber, & Voit, 2003)

374
375 The CE could only be calculated for the gelatin phase based on equation (3). Unfortunately, the
376 CE of the starch phase could not be calculated separately due to overlap of the characteristic
377 peaks of the starch and gelatin phase both present in the IPNs. Therefore, equation (3) is only
378 applicable for the gelatin phase present in the IPNs. It is important to emphasize that the CE
379 reflects a ratio between the amount of double bonds consumed upon crosslinking to the amount
380 initially present in the samples.

381



382
383 **Figure 2** 3D-plot representing the gelatin crosslinking efficiency (CE, z-axis) (%) of the various hydrogel films as a function of
384 the starch content (x-axis) and the degree of substitution of methacrylamide-modified gelatin (gel-MA) hydrogels (axis).

385

386 The CE values of the applied gelatin phase for the various hydrogel films are represented in
 387 figure 2. In addition to these results, table 2 represents the calculated amount of network points
 388 present in the gelatin phase taking into account the amount of photocrosslinkable MA side groups
 389 present in the network and the CE.

390 The results for the pure gelatin hydrogels are in good correlation with previous reported results
 391 for hydrogels crosslinked under similar conditions.(Salamon et al., 2014) However, the latter
 392 paper did not comprise a comparison of different DS of gel-MA. Figure 2 indicates an increasing
 393 CE with increasing DS for the hydrogels solely consisting of a gelatin phase (blue bars). This
 394 increase is observed until a maximum in CE is reached at a DS of 72%, since the crosslinking
 395 efficiencies for gel-MA 72% and 95% are in the same range. The trend of increasing CE with
 396 increasing DS can be anticipated as more crosslinkable side groups will be incorporated along the
 397 backbone for a higher DS (see table 2). Thus, more double bonds will be in closer proximity, and,
 398 therefore more likely to react upon photo-crosslinking. Moreover, the CE remains similar
 399 between the pure gelatin film compared to the IPNs with a 10% starch content (SP1 hydrogel
 400 samples in table 2). An increase in CE is observed, however, upon addition of a 20% starch
 401 phase. The latter phenomenon is anticipated to be the result of a more pronounced phase
 402 separation occurring between starch and gelatin present in the IPNs which is more likely to occur
 403 for the SP2 gelatin-starch IPNs as already highlighted in the previous section. This phase
 404 separation ensures the gelatin chains to exist in closer proximity despite the presence of an
 405 additional starch phase within the polymer network.

406

407

Table 2 Comparison of the amount of networks points in the gelatin phase with the amount of crosslinkable moieties in this gelatin phase for the various gelatin hydrogels samples developed as well as the interpenetrating networks based on gelatin and starch.

Code	mol MA moieties/ ml precursor solution	CE (%)	mol MA network points/ ml precursor solution
gel-MA 31%	1.19E-05	37	4.39E-06
gel-MA 72%	2.77E-05	65	1.82E-05
gel-MA 72% - SP1	2.49E-05	67	1.68E-05
gel-MA 72% - SP2	2.22E-05	92	2.04E-05
gel-MA 95%	3.66E-05	64	2.35E-05
gel-MA 95% - SP1	3.29E-05	79	2.59E-05
gel-MA 95% - SP2	2.93E-05	88	2.58E-05

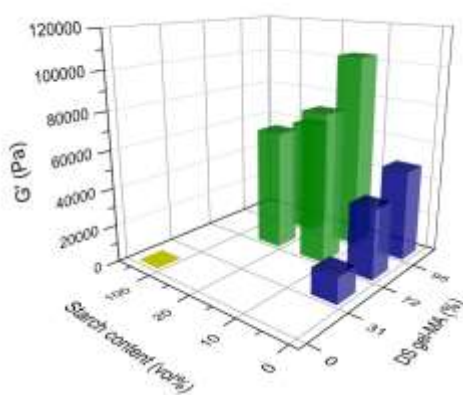
415

416 **m**
 ination of mechanical properties

417

418 Rheology was applied to examine the mechanical properties of the developed hydrogels. Polymer
419 materials typically exhibit visco-elastic behavior which implies that a recovery occurs at a certain
420 delay after deformation. As anticipated, an improvement in mechanical properties is observed for
421 more densely crosslinked hydrogels.(Hutson et al., 2011; Nichol et al., 2010; Van Den Bulcke et
422 al., 2000; Wang et al., 2014) This trend can be derived from figure 3 for the gel-MA and gel-MA
423 SP1 series along the y-axis: the storage modulus (G') increases with increasing DS of gel-MA.

424



425

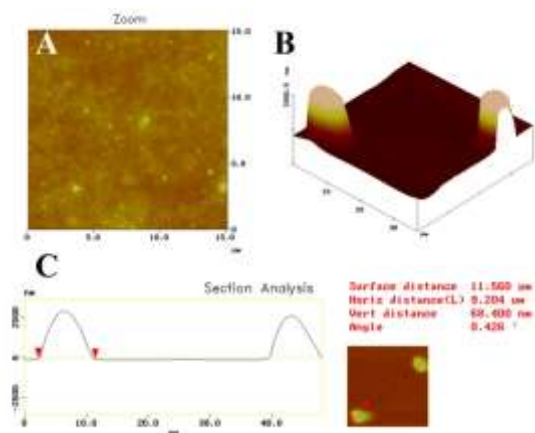
426 **Figure 3** 3D plot representing the storage modulus G' of the various hydrogels (z-axis) as a function of the starch content (%)
427 (x-axis) and the degree of substitution of methacrylamide-modified gelatin (gel-MA) (y-axis).

428 Although HR-MAS ^1H NMR spectroscopy indicated the highest CE for
429 gel-MA 72%, there is a lower absolute number of network points present compared to
430 gel-MA 95% (see last column table 2). Therefore, the hydrogel films consisting of gel-MA 95%
431 are characterized by a higher G' -value as these networks are more crosslinked. Moreover, G'
432 shifts to higher values for the IPNs with a starch-content of 10%. The mechanical properties are
433 thus improved upon introducing an additional starch phase in the gelatin network. For the IPNs
434 with 10% starch content (SP1), the trend along the y-axis remains similar: G' increases with
435 increasing DS of gel-MA. A more crosslinked gelatin phase thus results in improved mechanical
436 properties. Conversely, the IPNs with 20% starch content (SP2) again exhibit lower G' values
437 than the IPNs with 10% starch (SP1). It can be anticipated that the addition of a critical amount
438 of starch will result in a more pronounced phase separation, as already indicated above. In
439 addition, the gel fraction results complement the data and trends as derived from rheology.

440 **3.1.4. Topographical characterization**

441 The gelatin-starch IPNs were further investigated by AFM and IR mapping in order to study
442 relevant phase separation phenomena.(Dazzi et al., 2012; Ferrer, Sánchez, Ribelles, Colomer, &

443 Pradas, 2007) First of all, AFM is applied, a technique being part of the family of scanning probe
444 microscopes which scan across a surface monitoring probe-sample interactions. The
445 measurements were performed on spincoated gelatin/starch-solutions, since AFM experiments
446 require flat surfaces. In addition, spincoated gelatin and starch-solutions were also measured
447 separately as reference.
448



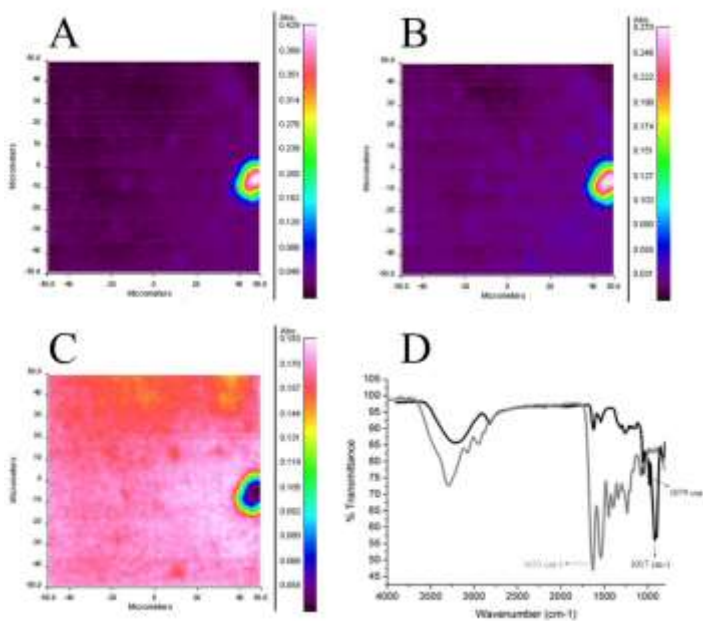
449
450 **Figure 4 A. Top view of methacrylamide-modified gelatin (gel-MA) B. 3D surface plot of 90% gel-MA + 10% starch-pentenoate**
451 **C. Section analysis of starch granules present in a gelatin matrix.**

452
453 The mixtures of gelatin and starch explicitly show smaller regions of phase-separated starch
454 granules being present adjacent to the globular domains of gelatin. These granules are separate
455 domains possessing a size of approximately 10 μm (figure 4).
456

457 In addition to AFM, the incorporation of starch in the gelatin matrix was also evaluated by means
458 of IR mapping of the characteristic wavenumbers of either gelatin (eg. 1633 cm⁻¹) or starch
459 (eg. 1017 and 1079 cm⁻¹). The results of the air-dried gel-MA 72%-SP1 hydrogel are depicted in
460 figure 5, together with the ATR-IR spectra of the starting materials. The results obtained from IR
461 mapping clearly confirm the phase separation occurring between gelatin and starch. A separate
462 starch domain was observed in the gelatin matrix exhibiting absorbance at the characteristic
463 wavenumbers corresponding with C-O bond stretching. Moreover, the size of this starch domain
464 is around 10 μm, which is in correlation with the AFM data (figure 4). Phase separation between
465 mixtures of gelatin and starch was already reported earlier.(Firoozmand et al., 2009; Firoozmand,
466 Murray, & Dickinson, 2012; Khomutov, Lashek, Ptitchkina, & Morris, 1995; Whitehouse,
467 Ashby, Abeysekera, & Robards, 1996) This phenomenon is mainly depending on the thermal
468 conditions, the carbohydrate molecular structure and the properties of the aqueous solution
469 including temperature, pH and ionic strength.(Firoozmand et al., 2012) Firoozmand et al. (2009)
470 also observed phase separation between gelatin and starch in high-sugar gelled systems

471 consisting of a constant gelatin content (7 wt%) and variable oxidized starch content (from 0 up
472 to 6 wt%).(Firoozmand et al., 2009) For this specific type of system, a microstructure could be
473 observed exhibiting both gelatin- and starch-rich regions with these regions ranging in size from
474 a few micrometers up to twenty micrometers observed via optical microscopy. However, it is
475 important to emphasize that the phase separation and the size of the domains was highly
476 dependent on the specific thermal treatment of the samples.

477
478



479
480 **Figure 5** IR spectroscopy data of a dry gel-MA 72%-SP1 starch hydrogel film, including an IR map depicting the absorbance at
481 **A.** 1017 cm^{-1} , **B.** 1079 cm^{-1} , **C.** 1633 cm^{-1} and **D.** the ATR-IR spectra of gel-MA (light grey) and starch-pentenoate (dark grey).

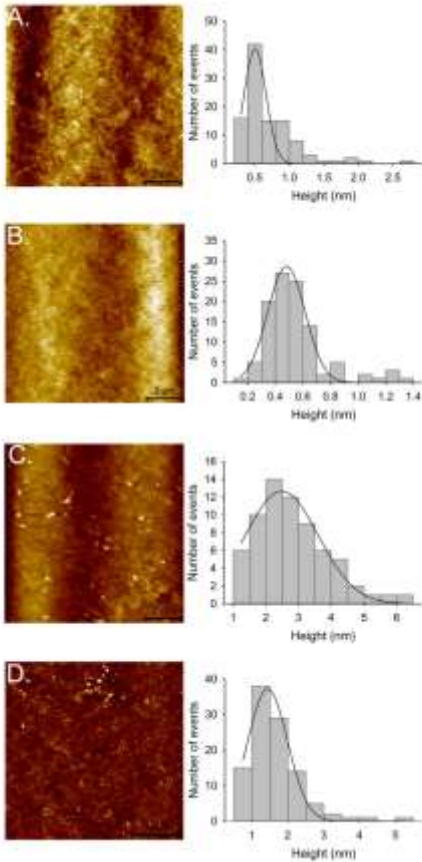
482 **3.2. Bioactive coating of gelatin hydrogels: aggrecan under investigation**

483 The application of cell-interactive ECM-based coatings is crucial when it comes to tissue
484 engineering, as these coatings can positively influence the cell growth(Altankov et al., 2000;
485 Franck et al., 2013; Heller et al., 2015; Shin, Jo, & Mikos, 2003).
486 In the present paper, aggrecan was selected as a component of the ECM to be applied on the
487 gelatin hydrogels. To the best of our knowledge, no data is yet reported on the application of an
488 aggrecan coating onto gelatin. Aggrecan is a major structural proteoglycan of the cartilage
489 extracellular matrix with a molecular mass higher than 2500 kDa.(Kiani, Chen, Wu, Yee, &
490 Yang, 2002) This molecule consists of numerous chondroitin and dermatan sulphate chains

491 attached to a core protein. In the present work, AFM and radiolabeling studies were performed in
492 order to determine the interaction between gelatin and aggrecan.

493 First, AFM was selected to examine the interaction between aggrecan and gelatin, as it allows
494 real-time imaging under liquid conditions, while providing a means to interrogate forces of
495 interaction at picoNewton resolution. The coated gelatin hydrogels were visualized by means of
496 tapping mode AFM before and after coating of the hydrogel surface with the proteoglycan (see
497 figure 6). At a concentration of 50 $\mu\text{g/ml}$ of aggrecan, no distinct features appear in the
498 topographic image of the coated hydrogel. Moreover, the height features detected are within the
499 same range as a non-coated gelatin hydrogel sample (see figures 6A and 6B). Thus, for this
500 concentration, no aggrecan can be detected on top of the gelatin hydrogels. The results from
501 figure 6C clearly show that features between 1.5 and 3 nm and even up to 6 nm are present on the
502 gelatin surface after aggrecan coating at a minimal concentration of 200 $\mu\text{g/ml}$. For a
503 concentration of 500 $\mu\text{g/ml}$, a high number of features sized between 1.5 and 2.5 nm can be
504 detected which indicates the presence of more aggrecan on the surface of the gelatin hydrogel.

505

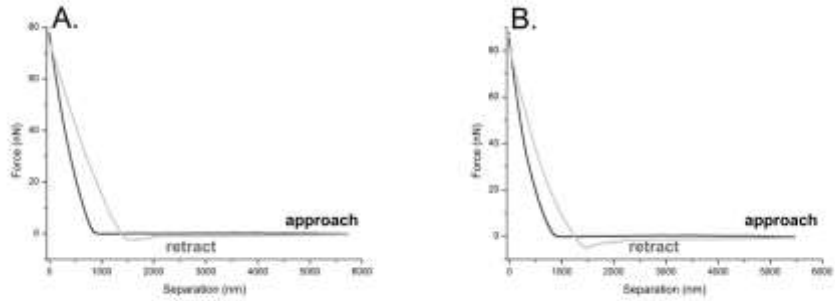


506
 507 **Figure 6** Topographic atomic force microscopy images of spincoated, crosslinked methacrylamide-modified gelatin sample **A.**
 508 **without** aggrecan-coating and at an aggrecan concentration of **B.** 50 $\mu\text{g/ml}$, **C.** 200 $\mu\text{g/ml}$, and **D.** 500 $\mu\text{g/ml}$. Images were
 509 obtained in liquid environment (PBS) at 20°C applying tapping mode.

510 Following topographic imaging of the surface, force spectroscopy experiments were performed to
 511 further characterize the gelatin-aggrecan affinity. For this reason, the operation mode was
 512 switched to contact mode and the AFM tip was functionalized with aggrecan. This procedure of
 513 tip functionalization via physical interactions allows dangling aggrecan molecules to be "pulled
 514 off" the surface that they are in contact with.(Florin et al., 1995) Figure 7 represents the force-
 515 distance curves obtained for the gel-MA samples and thus reflecting the adhesion force between
 516 aggrecan and gelatin. These forces of interaction between aggrecan and the hydrogel slightly
 517 increase from 0.97 to 1.25 nN with increasing DS of gel-MA. The forces detected are in the same
 518 range compared to the forces detected between proteins and biomaterial surfaces including

519 collagen and hyaluronic acid.(Donlon, Nordin, & Frankel, 2012; Herman-Bausier & Dufrene,
520 2016)

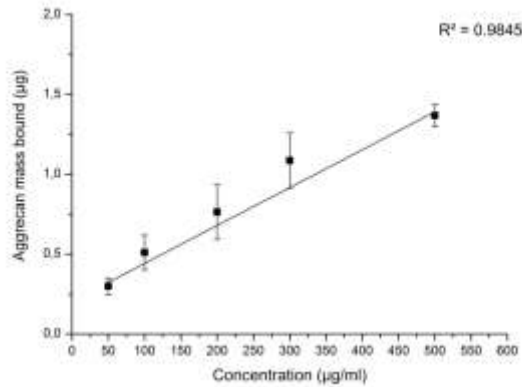
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557

558 **Figure 7** Force spectroscopy experiments of functionalized aggrecan-AFM tip absorbed on spincoated methacrylamide-
559 modified gelatin samples with a degree of substitution of A. 71 % , and B. 94 %

560 In a second part of the affinity study, radiolabeling experiments were performed enabling the
561 determination of the absolute mass of bound aggrecan. For these experiments, aggrecan was
562 radiolabeled with ^{125}I , and subsequently, a series of different concentrations of radiolabeled
563 aggrecan was coated on top of the gelatin hydrogels. The experiments were performed in
564 triplicate and the mean values and corresponding standard deviations are depicted in figure 8. The
565 results clearly indicate a dose-responsive signal which is nearly linear within the studied
566 concentration range from 50 to 500 $\mu\text{g/ml}$ aggrecan. It can be concluded that the radiolabeling
567 experiments enable characterization of the aggrecan/gelatin affinity at lower concentrations (i.e.
568 50 $\mu\text{g/ml}$) than liquid AFM which could only visualize concentrations starting from 200 $\mu\text{g/ml}$
569 aggrecan.



570

571 **Figure 8** Adsorbed aggrecan amount (µg) as a function of the aggrecan concentration (µg/ml) applied on methacrylamide-
 572 modified gelatin measured by radiolabeling experiments.

573 **4. Conclusions**

574 With the aim to investigate how the physico-chemical properties of biopolymer-based hydrogel
 575 films can be fine-tuned, hydrogel films were developed with varying chemical composition and
 576 degree of substitution of the functionalized gelatin. It can be concluded that the mechanical
 577 properties of the hydrogels can be fine-tuned depending on the degree of substitution of the
 578 methacrylamide-modified gelatin as well as the chemical composition (i.e. ratio gelatin/starch).
 579 The latter is reflected by the storage modulus of the developed materials which ranges from 14 to
 580 63 kPa. Furthermore, phase separation was observed for the IPNs as separated starch domains
 581 were present in the gelatin matrix. In addition, the present work also aimed at studying the
 582 affinity of aggrecan for gelatin. This affinity was successfully demonstrated via liquid atomic
 583 force microscopy and radiolabelling experiments. Thus, it can be concluded that gelatin-based
 584 hydrogels can be coated with aggrecan via physisorption. [In a forthcoming paper, an *in vitro* cell
 585 assay will be performed using human mesenchymal stem cells in order to evaluate the adipogenic
 586 as well as osteogenic differentiation potential of the hydrogels developed herein](#)

587 ~~In the subsequent ‘part B’ of this paper, all hydrogels developed will be evaluated for their
 588 potential to support adipose as well as osteogenic tissue regeneration in an *in vitro* approach
 589 based on human mesenchymal stem cells. In addition, the effectivity of a bioactive coating on top
 590 of the gelatin hydrogels films on these differentiation pathways will be assessed.~~

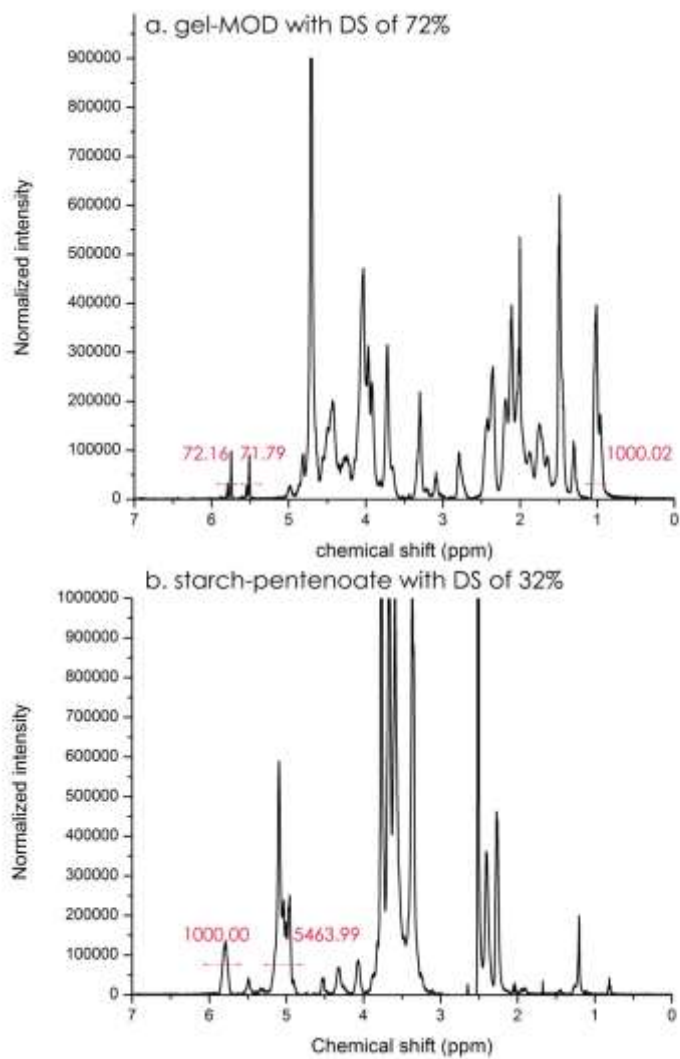
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603 Regeneration—ASCaffolds".
604

605 Supporting information



606

607

608

Figure S 1 ^1H NMR spectrum of a. gel-MA in D_2O at 40°C , b. starch-pentenoate in DMSO-d_6 at 60°C .

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